

# Enhancement Of Photovoltaic Properties Of CBD Grown Pr: Cds/Pbs Solar Cells

A. Mohamed Haroon Basha, K.K. Sivakumar, R. Chandramohan, N.Kavitha and R. Swaminathan

**Abstract:** With the advent of nanocrystalline based approach problems like performance degradation due to Pb diffusion in CdS/PbS solar cells can be solved. In this study an effort has been made to improve the performance of CdS/PdS thin film based devices, by doping with rare earth Pr. In this study undoped and Pr doped CdS thin films were prepared using Chemical Bath deposition and characterized for structural, optical and Morphological properties on pre cleaned glass and ITO coated glass plates. The PbS thin films were prepared system by chemical bath deposition over the CdS nano thin film layers grown over ITO glass plates and their photovoltaic properties were studied using Photocurrent–voltage measurements at AM 1.5 at an illumination of 100 mW/cm<sup>2</sup>. The results are reported. The increase of Pr concentration in the CdS layer has produced considerable enhancement in the microstructural and optical properties leading to a performance efficiency. The possible mechanism are also discussed.

**Index Terms:** CdS thin film, praseodymium doping, Pr doped CdS/PbS, structural, optical, photovoltaic properties

## 1. INTRODUCTION

Cadmium sulfide (CdS) is a well-known II-VI group semiconductor of much interest with high photoconductivity in the visible region, high electron affinity. The favorable electronic and electronic and optoelectronic properties of CdS nano thin films made this system interesting for sensors, solar cells, photo detectors, optical wave guides, nonlinear optical devices, and other opto electronic devices [1-3]. Though CdS is one of the most studied materials from past decades, its application as n-type window layer for high efficiency thin film hetero junction solar cells based on CdTe and Cu (InGa)Se<sub>2</sub> (CIGS offer many edges over other candidates of II VI family. CdS/PbS solar cells had shown significant promise, but the problem of lead diffusion into and CdS layer led to a long-term performance degradation and ultimately this hetero junction lost interest. However, the advent of nanocrystalline materials based approaches involving milder processing conditions, this problem can be easily avoided [5]. To improve the performance and efficiency of such thin film based systems, the properties of the layers should be modified. It has been experimentally proved that doping can be used as one of the possible ways to improve the electronic and optoelectronic properties of metal chalcogenide thin films. Many researchers have reported an enhancement in the optoelectronic properties of CdS thin film layers by doping with some cationic (In, Ni, Co) and anionic (Cl, Br) elements [6-10]. The photo generated electron–hole pairs are separated in the depletion region of the hetero junction creating a potential across the junction. If the junction is degraded then a stable performance could not be achieved.

Currently efforts have been devoted to the preparation of high quality CdS and PdS nanostructured thin films. Few low cost chemical techniques such as spray pyrolysis [12], successive ion layer adsorption and reaction (SILAR) [13], electrochemical growth method [14], and chemical bath deposition (CBD) [15] have exhibited amazing potential for preparing such thin films offering tailoring of surface properties. Among all of these methods, CBD is a simple technique to grow thin films that are uniform and can be easily scaled to industrial processings with ease. In this work undoped and Pr doped CdS thin films and PbS thin films were grown over glass substrates and their micro structural and absorption properties were studied and reported. For the preparation of hetero junction solar cells CdS were grown over ITO substrates of sheet resistance 10 ohms per square and the PbS thin films were grown over the CdS layers using CBD and their photovoltaic properties are reported.

## 2 PROCEDURE FOR EXPERIMENTATION

### 2.1 Chemicals Required

The typical CBD growth conditions of the CdS layer involves Cadmium Chloride (CdCl<sub>2</sub> 0.005 mol/L) and Ammonium Chloride (NH<sub>4</sub>Cl) in 100 ml deionized water at room temperature. The solution mixture was stirred and heated to 70 °C. After heating the solution for 5 minutes 0.04 mol/L of liquid ammonia solution was added and stirred for 5 minutes. To this solution 0.03 mol/L of thiourea was added subsequently and stirred continuously at the same temperature. This solution serves as the CBD bath. Pre cleaned glass slides were used as substrates and were placed vertically in the CBD bath for a period of 60 minutes. After the immersion of substrate for desirable time they were removed and cleaned with deionized water. The liquid ammonia combines with Cd<sup>2+</sup> ions and forms Cd(NH<sub>3</sub>)<sub>4</sub><sup>2+</sup> complex which releases Cd<sup>2+</sup> ions steadily. The thiourea releases S<sup>2-</sup> ions and acts as a sulfur source. These ions combine to form CdS layers over the surface provided. For the preparation of 1% and 3% Pr-doped CdS thin films the addition of 1% and 3% (0.005 mol/L) Praseodymium Chloride to the CBD bath before adding liquid ammonia and the experiment was repeated. For the preparation of solar cells, PbS thin film was deposited over CdS layer formed over ITO substrates (sheet resistance of ITO is 10 Ω/sq). A 50 ml solution of 0.15 M lead nitrate [Pb(NO<sub>3</sub>)<sub>2</sub>] and 0.1 M thiourea [SC(NH<sub>2</sub>)<sub>2</sub>] was used

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for the deposition of PbS. The alkalinity of the solution was set using 0.5 M sodium hydroxide [NaOH]. The CdS coated ITO substrate was vertically immersed into the solution and the beaker containing the reactive solution was kept in a water bath maintained at 60 °C for 1 hr.2.2 Final Stage

## 2.2 Characterization

Thickness of the prepared films was measured using an Alpha-step surface profiler. The structural properties of the grown layers were studied using X'pert PRO (PANalytical) X ray diffractometer employing CuK $\alpha$  radiation ( $k = 0.15405$  nm) in the  $2\theta$  range of 10–80°. Morphological studies were undertaken using Hitachi (S-3000H) scanning electron microscope. The optical studies were performed using Perkin Elmer Lambda 35 spectrophotometer. For the solar cell studies the Photocurrent–voltage measurements were performed using a Keithley 4200 semiconductor parameter analyzer under AM1.5G illumination at 100 mW/cm<sup>2</sup>.

## 3 RESULTS AND DISCUSSION

### 3.1 X-ray and Structural analysis

The X-ray diffraction patterns of undoped CdS, Pr doped CdS and PbS films respectively are presented in figure 1. The diffraction peak observed at 26.58° corresponds to the (002) plane of CdS wurtzite structure and matches the JCPDS card no. 41-1049. The XRD pattern of PbS also shows three characteristics peaks corresponding to (1 1 1), (2 0 0) and (2 2 0) orientations, in agreement with the standard JCPDS card No. 00-0050592 of cubic rock salt (NaCl) type structure [16]. Using the usual expressions the crystallite size, micro strain and the stacking fault probability was calculated by measuring the peak shift with the standard value [17]. The intensity of the prominent peak (002) is found to increase as the film doping concentration increases. Structural parameters of the CdS and PbS thin films were also calculated and shown in Table 1. As the CdS film thickness increases with increase of Pr doping concentration, the reduction in microstrain and dislocation density were observed which might have lead to the relaxation of the stress in the films. This relaxation produces break between the substrate and the layer as the Pr doping concentration increases and hence the stacking fault decreases as given in the Table 2.

### 3.2 Morphological and compositional analysis

Figure 2 (a-c) shows the morphology of Pr: CdS thin films. These SEM micrographs clearly reveal the homogeneity of the CdS thin films. The surface morphology is found to be affected by the Pr doping slightly. It is also apparent from the SEM image shown in Fig. 2(a) that the undoped film has morphology formed using nearly spherical grains. Figures 2(b-c) show the SEM images of the films prepared after 1% Pr doping which indicate uniform covering of the substrate with the spherical shaped grains. At 3% Pr doping concentration also the morphology do not change and the uniform coverage of the grains are observed revealing the high quality of the nano thin film layers facilitating the mobility and diffusion of surface atoms [18].

### 3.3 Compositional analysis (EDX)

The elemental composition of chemical bath deposited CdS thin films were also studied using the EDX. Figure 2 (d, e) shows the typical EDX spectra of the undoped and 3% Pr-

doped CdS thin films. The spectrum shows Cd and S peaks, for undoped and Pr peak confirming the presence of Cadmium (Cd), sulfur (S) and (Pr) praseodymium in the doped sample. The films showed nearly stoichiometric composition. The EDX spectrum showed extra peaks corresponding to substrate.

### 3.4 Optical properties

The changes in optical transmittance (T) with the incident wavelength ( $\lambda$ ) for the CdS thin films prepared using different Pr doping concentration are study using the transmission spectra shown in Fig. 3 in the wavelength region 400–1100 nm. It is clear that all films are highly transparent in the visible wavelength range and showed sharp ultraviolet absorption edge at approximately near 540 nm. The presence of multiple interference fringes in the transmittance spectrum denotes that the surface is smooth. In Fig. 3 the inset shows transmittance spectrum of PbS film and it shows a value of transmittance around 35%. The optical absorption spectra of undoped and Pr doped thin films are also recorded at room temperature from 400 to 1000 nm as shown in Fig. 4. This absorption spectrum shows that the absorption is high in the blue region of visible spectrum. The Fig. 4 inset shows absorption spectrum of PbS film and it shows a higher absorption in the visible region. Using the usual Tauc's relation the band gap was evaluated. Figure 5 shows the plot of  $h\nu$  vs  $(\alpha h\nu)^2$  of the films deposited at different Pr doping concentration. The calculated energy gap ( $E_g$ ) is found to decrease from 2.48 eV to 2.43 eV with the increase of Pr doping concentration from 0% to 3%, showing the crystallinity improvement with the densely packing of atoms [19]. The inset of figure 5 shows the direct energy gap of PbS thin film deposited by chemical bath deposition method. The obtained band gap value of PbS is 1.54 eV which agrees with those reported [20]. Optical constants of CdS thin films such as refractive index ( $n$ ) and extinction coefficient ( $k$ ) were calculate using the reported relations [18]. Fig. 6 and Fig. 7 show the variation of both  $n$  and  $k$  of CdS films deposited with different Pr doping concentration. The refractive index of the films decreased with the increase of doping concentration. The increase of Pr may impede the propagation of light through the film [21]. The extinction coefficient of the films varied in the range from 0.24 to 0.30 [22,23,24].

## 5 FIGURES AND TABLES

**Table 1** Thickness and structural parameters of CdS, Pr-CdS and PbS thin films

Pr doping in CdS film	Thickness (μm)	Crystal lite size (nm)	Dislocation density ( $\delta \times 10^{15}$ lines. m <sup>-2</sup> )	Micro Strain ( $\epsilon \times 10^{-3}$ lines <sup>-1</sup> . m <sup>-4</sup> )	Stacking fault probability ( $\gamma$ ) $\times 10^{-4}$
0%	0.61	28	1.27	4.3	2.9
1%	0.65	31	1.05	3.9	2.4
3%	0.70	37	0.78	3.2	1.8
PbS	0.88	43	0.54	2.3	1.3

**Table 2** Parameters of the solar cell structures in Fig. 8

parameters	Pr doping in CdS film

	Pr 0%	Pr 1%	Pr 3%
$V_{oc}(V)$	0.31	0.32	0.33
$I_{sc} (mA/cm^2)$	7.08	7.91	8.72
FF	0.35	0.36	0.38
$\eta$ (%)	1.14	1.32	1.66

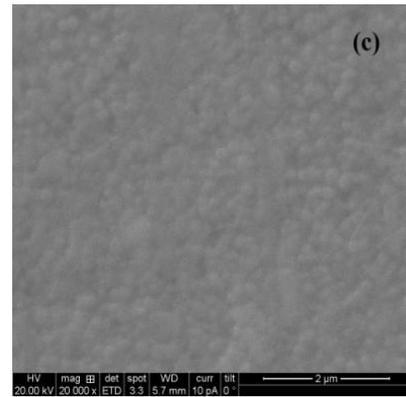


Figure 2 (c) 3% Pr doped CdS

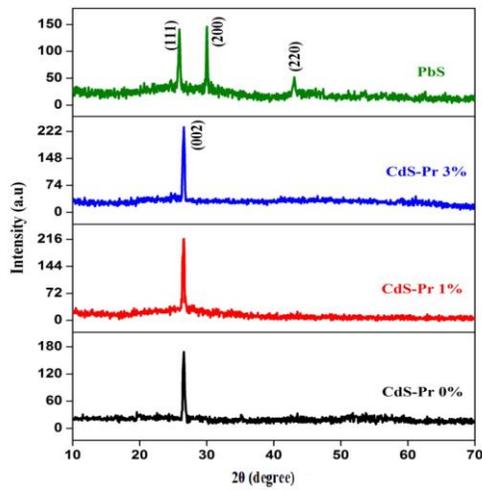


Figure 1 XRD patterns of Cds and PbS thin films

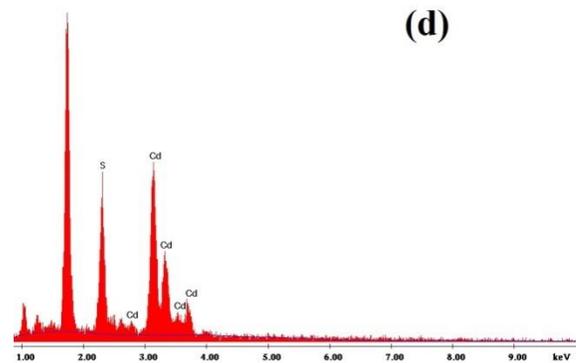


Figure 2 (d) 0% Nd doped EDX spectrum

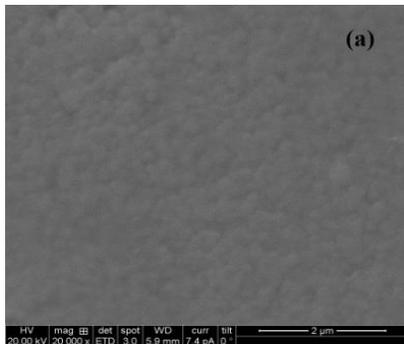


Figure 2 (a) SEM images of CdS thin films undoped 0%

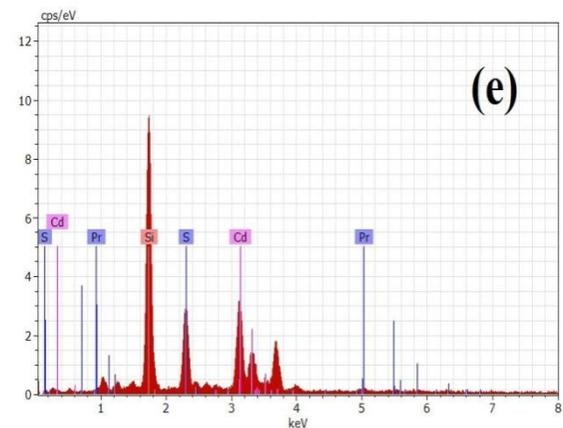


Figure 2 (e) 3% Pr doped EDX spectrum

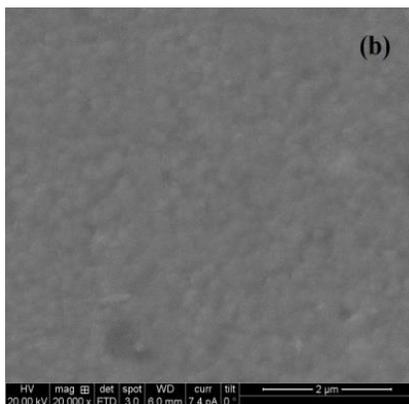
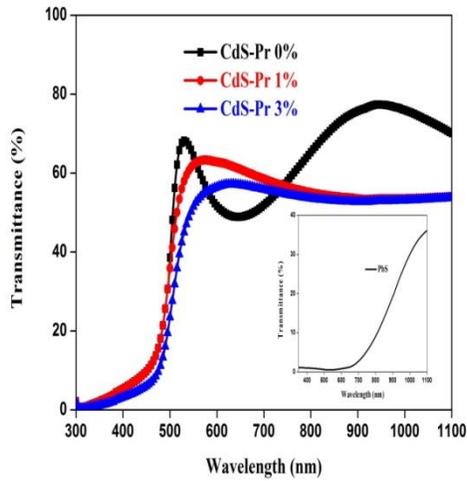
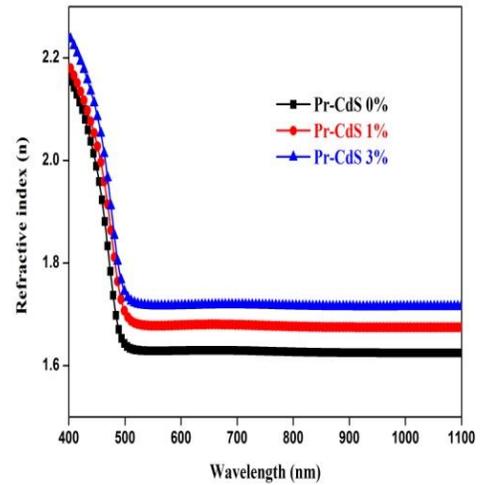


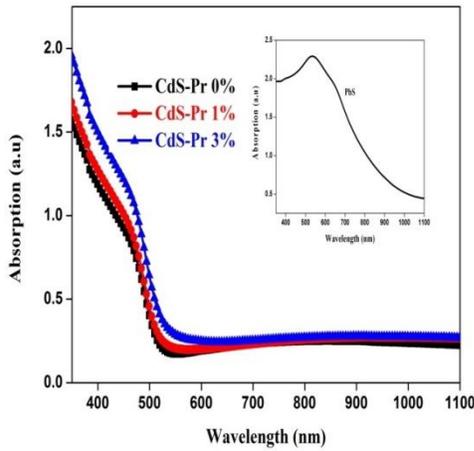
Figure 2 (b) SEM images of CdS thin films 0%



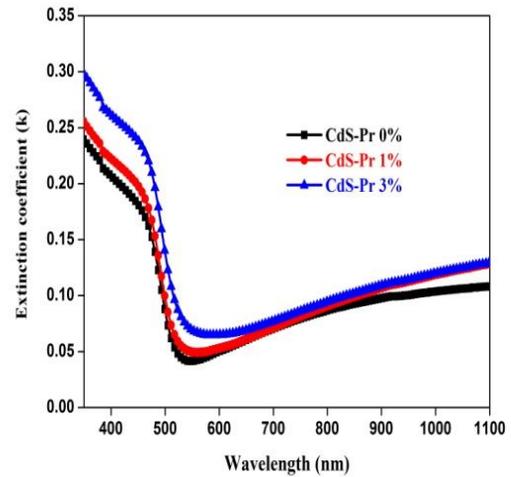
**Figure 3** Optical transmission spectra of CdS and inset shows PbS thin films



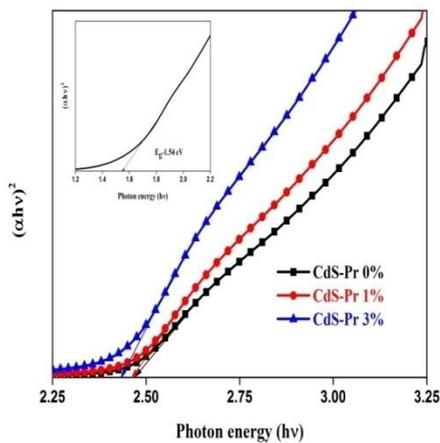
**Fig. 6** Refractive index (n) of CdS thin films prepared at different Pr doping concentration



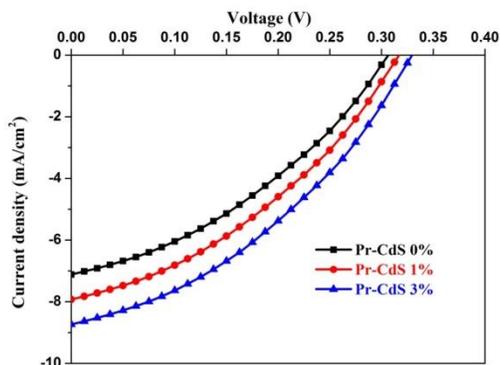
**Figure 4** Optical absorption spectra of CdS and inset shows PbS thin films



**Fig. 7** Extinction coefficient of CdS thin films prepared at different Pr doping concentration



**Fig. 5**  $(\alpha hv)^2$  vs  $(hv)$  plot for CdS and inset shows PbS thin films



**Figure 8** I-V curves for the solar cells ITO/ n-Pr-CdS/ p-PbS/Al structure

#### 4 CONCLUSION

Pr-CdS, PbS and CdS/PbS heterostructures were successfully prepared using simple chemical bath deposition method. Praseodymium doping to n-CdS layer has shown improved photovoltaic properties of the ITO/n-Pr-CdS/PbS/Al heterostructure. XRD confirmed the formation of hexagonal phase with a preferred orientation in the (0 0 2) plane. The PbS film showed cubic rock salt structure with (200) preferred orientation. Crystallite size of the Pr doped CdS is found to increase with Pr concentration, and it is found to be 37 nm for the 3% Pr concentration. The surface uniformity was excellent for all the films and Pr doping retained surface uniformity and improved mobility for photo generated carriers. Optical studies revealed that the optical transmittance and energy band gap decreased with the increase of Pr doping concentration. The solar conversion efficiency of the fabricated solar cells is found to increase from 1.14 up to 1.66 %, as Pr doping concentration is increased from 0% to 3% and the possible mechanism are discussed.

#### ACKNOWLEDGMENT

The authors would like to thank their Management and Hon Vice Chancellor and Principals of their concerned University and college for their constant encouragement and financial assistance.

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